

동력시스템공학회지 제28권 제6호 pp. 40-47 2024년 12월 ISSN 2713-8429(Print) ISSN 2713-8437(Online) Journal of Power System Engineering https://doi.org/10.9726/kspse.2024.28,6,040 Vol. 28, No. 6, pp. 40-47, December 2024

Fatigue Limit Evaluation for Small Crack in Ni 12% STS316L

Jung-Kyu Lee* †

(Received 30 September 2024, Revision received 12 November 2024, Accepted 12 November 2024)

Abstract: In this study, the fatigue limit and threshold stress intensity factor of Ni 12% STS316L were obtained. The fatigue limit and threshold stress intensity factors of small cracks were also evaluated. The fatigue limit of crack specimens showed good agreement between experimental and calculated results. Furthermore, the crack size evaluated under the maximum operating pressure (87.5 MPa) of the hydrogen storage tank confirmed the safety of the input/output piping system.

Key Words: Fatigue Limit, Threshold Stress Intensity Factor, Input/output Piping System, Operating Pressure

1. Introduction

Since the American electric vehicle and clean energy company, Tesla, was founded in 2003, the automotive market has transitioned from traditional internal combustion engines to eco-friendly vehicles. In the past decades, research and development based on renewable energy has been actively carried out as a solution to global warming.¹⁻⁶⁾ Hydrogen fuel cell vehicles are creating a differentiated market from conventional electric vehicles.^{7,8)} Theoretically, as the hydrogen charging capacity of the vehicles is increased, the driving distance can be lengthened. The hydrogen storage tank is made by winding carbon fibers on high-strength plastic composites. Khan et al.⁹⁾ simulated the temperature rise of the tank wall during hydrogen filling of a carbon fiber reinforced plastic tank. The results were compared with the experimental data of the Japan Automobile Research Institute (JARI). The hydrogen storage tank

* † Jung-Kyu Lee(https://orcid.org/0009-0000-5196-9899) : Technical Advisory, Fine Technplogy Co., Ltd.

E-mail : pknua@hanmail.net, Tel : 051-629-6289

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is about 60% lighter and more than 10 times stronger than the conventional metal fuel tank. The hydrogen storage tank has a working pressure of 70 MPa, a maximum working pressure of 87.5 MPa, and a working temperature of -40 to 85° C.¹⁰ Regulators, manifolds, and solenoid valves of hydrogen storage tanks use Ni 12% STS316L to minimize the effects of corrosion and hydrogen embrittlement.¹¹ As these parts are subjected to frequent temperature and pressure fluctuations during operation, the presence of micro cracks can lead to catastrophic failures resulting in human fatalities and and significant economic losses.

The micro crack problem is not established a small-scale nonlinear region assumption, while it is inherently nonlinear problem. Haddad et al.¹²⁾ proposed an evaluation equation for the crack length dependence of the threshold stress intensity factor (ΔK_{th}) based on the sum of crack length (l) and micro crack length (l_0) . Subsequently, Tange et al.¹³⁾ modified Haddad's equation to develop a more convenient expression. Meanwhile, Ando et al.¹⁴⁻¹⁶⁾ introduced a threshold stress intensity factor and fatigue limit evaluation equation for fatigue cracks

that incorporates the crack tip process zone, enabling a unified treatment of fracture mode, including brittle fracture, fatigue fracture, hydrogen embrittlement, and stress corrosion cracking. This equation demonstrated accurate prediction of the fatigue limit.

This study aimed to assess the safety of STS316L under the working stress conditions. Initially, the fatigue limit ($\Delta \sigma_w$) of smooth specimen and the threshold stress intensity factor ($\Delta K_{th(l)}$) of the large crack were determined. Subsequently, the fatigue limit ($\Delta \sigma_{wc}$) of a cracked specimen and the threshold stress intensity factor ($\Delta K_{th(s)}$) of the small crack were evaluated. Furthermore, by estimating the crack size based on the maximum working pressure (87.5 MPa) and the fatigue limit reduction ratio of the hydrogen storage tank, and the safety of STS316L was confirmed.

2. Experimental method

2.1 Material

The material used in this study is austenitic stainless steel STS316L for the piping of hydrogen shows the chemical storage tanks. Table 1 compositions of the material, and Table 2 shows the mechanical properties. Fig. 1(a) illustrates a CT specimen with a thickness of 12.5 mm used to determine $\Delta K_{th(l)}$, while Fig. 1(b) shows a smooth specimen with dimensions of 10 mm width, 124 mm length, and 4 mm thickness used to obtain $\Delta \sigma_w$. Fig. 2 shows the dimensions of the semi-elliptical slit introduced into the crack specimen. To accurately evaluate the initiation and growth of a small crack, a very small crack is required. However, making such a small slit using electric discharge machining (EDM) was very difficult. Consequently, the dimensions were decided as shown in Fig. 2. The semi-elliptical slit was subsequently introduced

Table 1	Chemical	compositions	of	test	material
	(wt.%)				

С	Si	Mn	Р	S	Ni	Cr	Mo	Co
0.01	0.67	1.19	0.035	0.001	12.14	17.41	2.05	0.21

Table 2 Mechanical properties of test material

Yield strength	Tensile strength	Elongation	Hardness
(MPa)	(MPa)	(%)	(HRBW)
313	560	49	81

at the center of the smooth specimen through the EDM process. The aspect ratios (As=a/c) of the semi-elliptical slit are 1.0 and 0.4, where, a is the crack depth, and c is a half of the crack length. Depth a is varied as 0.2, 0.4, and 0.5 mm at As=1.0, and 0.1, 0.2, and 0.3 mm at As=0.4.



Fig. 1 (a) CT specimen, and (b) Smooth specimen for fatigue test (unit : mm)

0.1mm	<i>a/c</i> (mm/mm)		
1 2c 1	As=1.0	As=0.4	
C +	0.2/0.2	0.1/0.25	
at ey	0.4/0.4	0.2/0.5	
A	0.5/0.5	0.3/0.75	

Fig. 2 Shape and dimension of artificial surface defects

The aperture width of the slit was approximately 0.1 mm. To eliminate work-hardening after introducing the semi-elliptical slit, it was water-cooled after maintaining for 20 min at 1,050°C of Ar atmosphere.

2.2 Fatigue test method

The threshold stress intensity factor $(\Delta K_{th(l)})$ of the large crack and the fatigue limit $(\Delta \sigma_w)$ of the smooth specimen were investigated. The fatigue test was conducted at room temperature using a fatigue testing system (nominal dynamic load rating: ±100 kN). The experiment for $\Delta K_{th(l)}$ and $\Delta \sigma_w$ was carried out according to fatigue test method of Korean standard (KS B ISO 12108). $\Delta K_{th(l)}$ was estimated by the ΔK -decreasing test for long crack, and $\Delta \sigma_w$ was estimated by constant stress amplitudes test. The fatigue tests were performed with a stress ratio R=0.1. All the tests were performed at a frequency of f=20 Hz. The fatigue limit was defined as the maximum stress amplitude at which the specimen could endure 5×106 cycles.

2.3 Evaluation method

This study used Eq. (1), which was proposed by Ando et al.14) This equation describes the dependence of the threshold stress intensity factor range (ΔK_{th}) on crack length when an existing crack in an infinite plate propagates under fatigue stress.

$$\Delta K_{th} = 2\beta \Delta \sigma_w \sqrt{\frac{a}{\pi}} \cos^{-1} \left[\left\{ \frac{\pi}{8\beta^2 a} \left(\frac{\Delta K_{th(l)}}{\Delta \sigma_w} \right)^2 + 1 \right\}^{-1} \right]$$
(1)

Where a is the crack depth, $\Delta K_{th(l)}$ is the threshold stress intensity factor range for the large crack, $\Delta \sigma_w$ is the fatigue limit of the smooth specimen, β is a function of ϕ in Fig. 2 and a shape factor obtained by the Newman-Raju equation.¹⁷⁾ In contrast, the fatigue limit ($\Delta \sigma_{wc}$) of the cracked specimen can be evaluated using Eq. (2).

$$\Delta K_{th} = \Delta \sigma_{wc} \sqrt{\pi a} \tag{2}$$

Eqs. (1)~(2) are used to determine ΔK_{th} and $\Delta \sigma_{wc}$ for the deepest part of the crack (point A) in the cracked specimen to which bending stress is applied. To determine these values on the outermost surface (point C), a can be replaced with c in Eqs. (1) and (2).

Results and Discussion

To obtain $\Delta K_{th(l)}$, a K-decreasing test was conducted according to the KS standard using three compact tension (CT) specimens. Fig. 3 shows the results obtained from the experiment. $\Delta K_{th(l)}$ represents the ΔK value where da/dN approach zero. While typically defined as the ΔK corresponding to 10^{-8} mm/cycle for most materials, in this study, da/dN converged to zero (0) at 4.1×10^{-7} mm/cycle. This ΔK value was determined as the threshold stress intensity factor ($\Delta K_{th(l)}$) for the large crack, equaling 6.3 MPa·m^{0.5}.

Fig. 4 shows the S-N curve for determining the fatigue limit $(\Delta \sigma_w)$ of the smooth specimen. The



Fig. 3 Relationship between fatigue crack growth rate and stress intensity factor



Fig. 4 S-N curve for determining the fatigue limit of a smooth specimen

arrow symbol (\rightarrow) indicates specimens that did not fracture after 5×10⁶ cycle. Based on this data, the fatigue limit $(\Delta \sigma_w)$ for STS316L was determined to be 285 MPa.

Fig. 5 shows the relationship between fatigue limit ($\Delta \sigma_{wc}$) and crack depth for two aspect ratios: is As=1.0 (Fig. 5(a)) and As=0.4 (Fig. 5(b)). Solid circles () show fractured specimens, while open circles (O) indicate unfractured ones. The dotted line, representing the calculated fatigue limit curve as a function of crack depth, is obtained from Eq. 2. While the straight line in the figure denotes the fatigue limit ($\Delta \sigma_w$) of the smooth specimen, it's evident that the fatigue limit decreases with increasing crack depth. The slope of this decrease corresponds to the threshold stress intensity factor of the large crack. Notably, the reduction in fatigue limit is more pronounced for As=0.4. In the small As, the fatigue crack propagates in the depth direction, because the surface crack is large. When the crack depth increases to some extent, the surface crack begins to propagate. The experiment results were in good agreement with the calculation results using Eq. 2. The dotted lines in Figs. 4(a) and 4(b) represent fatigue limit reduction ratios of 10%, 15%, 25%, and 50%. Additionally, a dotted line indicating a 71% fatigue limit reduction, corresponding to the maximum working pressure of the hydrogen storage tank (87.5 MPa), was also shown.

Fig. 6 shows the relationship between crack depth and fatigue limit reduction ratio for various



Fig. 5 Fatigue limit according to aspect ratio. (a) As=1.0, (b) As=0.4



Fig. 6 Crack size as a function of the fatigue limit reduction ratio for each aspect ratio

As. Data points for As=1.0 and 0.4 were obtained from Fig. 4, while those for As=0.8 and 0.6 were calculated using Eq. 1. The results indicate that crack depth is influenced by both fatigue limit reduction ratio and

As. For a given fatigue limit reduction ratio, crack depth decreases as As decreases. Specifically, the crack depth for As=0.4 is approximately half that of As=1.0. Conversely, at a fixed As, crack depth increases with increasing fatigue limit reduction ratio. It's noteworthy that crack depths for all As remain relatively small up to a 25% fatigue limit reduction ratio. The fatigue limit reduction ratio of 50% was about 12 times of 10%, about 7.4 times of 15%, and about 3.6 times of 25%. That is, the crack lengths for fatigue limit reduction ratio of 50% were 0.983 mm, 0.783 mm, 0.6166 mm, and 0.485 mm for As=1.0, 0.8, 1.6, and 0.46, and As=0.4, respectively. For a 71% fatigue limit reduction ratio, corresponding to the maximum working pressure of 87.5 MPa, crack depths were 2.298 mm, 1.875 mm, 1.504 mm, and 1.188 mm for As=1.0, 0.8, 0.6, and 0.4, respectively. These depths represent 76.6%, 62.5%, 50.1%, and 39.6% of the 3 mm thick material used in this study. At this 71% reduction, calculated crack lengths were 2.298 mm, 2.344 mm, 2.507 mm, and 2.970 mm for As = 1.0, 0.8, 0.6, and 0.4, respectively.

Rummel et al.¹⁸⁾ investigated the detection ratio of semi-elliptical fatigue crack. They used ultrasonic nondestructive testing under optimal indoor conditions. Crack with a detection probability of 100% was dimensions of 2c=12 mm and a=4 mm. For a detection probability of 50%, the dimensions were 2c=1.2 mm and a=0.26 mm. The minimum detectable crack size was 0.17 mm in both depth and length. Recently, with the advancement of non-destructive technology, Ochiai et al.19) measured stress corrosion cracks with a depth of 0.4 mm using laser ultrasonic nondestructive testing. Consequently, when employing laser ultrasonic



Fig. 7 Fracture surface by fatigue

nondestructive testing, cracks that correspond to 50% and 71% of the fatigue limit reduction ratio in this study are 100% detectable, ensuring safety.

Fig. 7 shows a fatigue fracture surface, with arrows indicating the direction of fatigue crack propagation.

Crack occured at the tip of an artificial defect made by EDM. Numerous ratchet marks are observed at the crack tip, which are caused by the concentration of stress at the artificial defect's tip. These ratchet marks are fatigue indicators, as described by Becker et al.²⁰⁾

Fig. 8 shows the stress intensity factor (ΔK_{ap}) and the threshold stress intensity factor $(\Delta K_{th(s)})$ as functions of crack depth. ΔK_{ap} represents the stress intensity factor under applied stress, while $\Delta K_{th(s)}$ denotes the threshold stress intensity factor for the small crack. Figs. 8(a) and (b) show the results for aspect ratios As=0.1 and 0.4, respectively, with calculated using Eq. 1. The subscripts A and C denote crack depth and crack length, respectively. The values ${\it \Delta}K_{ap(a)}$ and ${\it \Delta}K_{ap(c)}$ obtained from the Newman-Raju equation correspond to crack depth and crack length, respectively. In Fig. 7(a), where As=0.1 and a=c, there is no significant difference between $\Delta K_{ap(a)}$ and $\Delta K_{ap(c)}$ and corresponding $\Delta K_{th(s)A}$ and $\Delta K_{th(s)C}$. However, the value of c is slightly larger. This difference was obtained because the equations for ΔK_{ap} and $\Delta K_{th(s)}$ use 0° for the surface crack and 90° for the depth crack. Both ΔK_{ap} and $\Delta K_{th(s)}$ increase with crack growth but do not intersect. An increase in ΔK_{ap} indicates that

the fatigue crack is propagating. As the fatigue crack propagates, $\Delta K_{th(s)}$ approaches $\Delta K_{th(l)}$. Fig. 8(b) shows that for an aspect ratio As=0.4, the crack depth (a) is larger than the crack length (c). This is because of the larger discrepancy due to the difference angles used for surface and depth cracks (0°for surface and 90°for depth). Consequently, $\Delta K_{ap(a)}$ is larger than $\Delta K_{ap(c)}$. This is consistent with the phenomenon that a smaller the As leads to faster propagation of the fatigue crack in the depth direction. In other words, as the crack propagates in depth and approaches the surface crack length, the surface crack length starts to propagate as well. It can be observed that $\Delta K_{ap(a)}$ increases as As decreases.



Fig. 8 Changes in stress intensity factor and threshold stress intensity factor with respect to aspect ratio. (a) As=1.0, (b) As=0.4

Conclusions

In this study, the threshold stress intensity factor and fatigue limit of Ni 12% STS316L were determined. The fatigue limit ($\Delta \sigma_{wc}$) and the threshold stress intensity factor ($\Delta K_{th(s)}$) of the small crack was evaluated. Additionally, the crack size was assessed at the maximum working pressure of 87.5 MPa, and ensuring the safety of the input/output piping system.

1) The fatigue limit ($\Delta \sigma_{wc}$) for crack specimens was calculated by using $\Delta K_{th(l)}$ and $\Delta \sigma_w$. $\Delta \sigma_{wc}$ decreased as the crack depth increased, with smaller values observed for smaller aspect ratios (As). As the crack grew, $\Delta K_{th(s)}$ approached $\Delta K_{th(l)}$. The experimentally determined $\Delta \sigma_{wc}$ showed good agreement with the calculated $\Delta \sigma_{wc}$.

2) For As=0.4, the crack depth corresponding to the fatigue limit reduction ratio was approximately half that of As=1.0. The cracks with a 50% reduction ratio were found to be approximately 3.6 times larger than those with a 25% reduction ratio. At the same fatigue limit reduction ratio, cracks with smaller As were evaluated as smaller and safer.

3) The cracks with a fatigue limit reduction ratio of 50% and 71% (at the maximum working pressure of 87.5 MPa) can be detected with 100% accuracy, and the safety of the piping system was ensured.

Author contributions

J. K. Lee; Conceptualization, Data curation, Investigation, Methodology, Resources, Writing-review & editing.

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